

"Gilberto Vlaic" XVII School on Synchrotron Radiation: Fundamentals, Methods and Applications

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# Synchrotron radiation-based X-ray techniques & cultural heritage objects: an integrated multi-material, multi-technique and multi-scale approach to study their composition and evolution

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#### SR-based X-ray methods for cultural heritage objects\*

Cultural heritage objects: heterogeneous and composite systems, in the most of cases composed of multiple layers, whose thickness can achieve values of a few micrometers.



Subject to chemical transformations, often involving changes of the oxidation state of elements and formation of polymorphs.

SR-based X-ray methods (imaging/mapping and single-point analysis mode): possibility of obtaining information about the chemical nature and distribution of different phases down to the submicrometer scale length.

a) micro-X-ray fluorescence (µ-XRF) for elemental microanalysis down to the sub-ppm level.

b) **micro-X-ray absorption spectroscopy (µ-XAS)** for probing the local chemical environment (oxidation state, coordination numbers, site symmetry and distortion, bond distances) of selected elements; it can be equally applied on amorphous or crystalline materials.

c) **micro-X-ray diffraction (µ-XRD)** for obtaining long range order information about the presence and nature of crystalline phases.

\* M. Cotte et al., Accounts of chemical research 43 (2010) 705-714; L. Bertrand et al., Appl. Phys. A 106 (2012) 377–396; K. Janssens et al., Annu. Rev. Anal. Chem. 6 (2013) 399–425; K. Janssens et al., Top. Curr. Chem. 374(6) (2016), doi:10.1007/s41061-016-0079-2; M. Cotte et al., JAAS 32 (2017) 477–493; V. Gonzalez et al., Chem. Eur. J. 26 (2020) 1703 –1719; S. Quartieri, Synchrotron Radiation in Art, Archaelogy and Cultural Heritage. In Synchrotron Radiation - Basics, Methods and Applications (S. Mobilio, F. Boscehrini, C. Meneghini Eds.), Springer (2015), pp. 677-695.

# Why SR for X-ray diffraction and spectroscopy?

➢ high X-ray flux. Fundamental requirement to obtain high signal-tonoise data in a reasonable time frame; parameter particularly critical if the element of interest is at low concentration in a sample. Synchrotron sources provide X-rays of five or more orders of magnitude greater flux than conventional laboratory X-ray sources.

Broad spectral range at uniformly high flux. A typical XAS/XRF spectrum covers about 1000 eV. Tunable monochromators with appropriate d-spacings can be used to scan through a broad range of energy; thus, it is possible select the most appropriate energy range for an experiment.

High stability in flux, energy and beam position.

## XANES/XRF/XRD experiments



# XANES/XRF mapping/imaging experiments



#### SR for Cultural heritage materials: further readings

Handbook of Cultural Heritage Analysis

Handbook of Cultural Heritage Analysis pp 45–67 Cite as

Home > Handbook of Cultural Heritage Analysis > Reference work entry

#### X-Ray Absorption Spectroscopy (XAS) Applied to Cultural Heritage

Francesco D'Acapito

In: D'Amico, S., Venuti, V. (eds) Handbook of Cultural Heritage Analysis. Springer, Cham. https://doi.org/10.1007/978-3-030-60016-7\_4



Home > Synchrotron Light Sources and Free-Electron Lasers > Reference work entry

#### Using Synchrotron Radiation for Characterization of Cultural Heritage Materials

Koen Janssens <sup>™</sup> & Marine Cotte

In: Jaeschke, E., Khan, S., Schneider, J., Hastings, J. (eds) Synchrotron Light Sources and Free-Electron Lasers. Springer, Cham. https://doi.org/10.1007/978-3-030-23201-6\_78

## **Historical glasses**

JPhys Photonics

#### **PAPER • OPEN ACCESS**

Unravelling the role of iron and manganese oxides in colouring Late Antique glass by micro-XANES and micro-XRF spectroscopies

To cite this article: Francesca Gherardi et al 2024 J. Phys. Photonics 6 025001



#### Manufacturing process of Atlantic tradition glass\*

- The combined presence of iron (Fe) and manganese (Mn) produces glasses with a wide range of colours (from green to amber to purple).
- The colour depends on the relative amounts of Mn and Fe in different oxidation states



#### experimental natron-type glass

#### $Fe(II)_{yellowish} + Mn(III)_{purplish} \rightleftharpoons Fe(III)_{bluish} + Mn(II)_{yellowish}$

Study of lab-prepared/experimental natron-type glass and archaeological glass from Tintagel and Whithorn (UK) (5th–7th century) by Fe and Mn K-edge XANES/XRF spectroscopy combined with UV-Visible spectroscopy and PCA to evaluate: (1) the influence of different glass production parameters; (2) the variation of Fe and Mn oxidation states; (3) the optical character.

#### Archaeological glass (Tintagel/Whithorn, 5th-7th century)



\*F. Gheradi et al., J. Phys. Photonics 6 (2024) 025001, https://doi.org/10.1088/2515-7647/ad2259

#### Manufacturing process of Atlantic tradition glass\*



ID21 beamline

\*F. Gheradi et al., J. Phys. Photonics 6 (2024) 025001, https://doi.org/10.1088/2515-7647/ad2259

#### Manufacturing process of Atlantic tradition glass\*



\*F. Gheradi et al., J. Phys. Photonics 6 (2024) 025001, https://doi.org/10.1088/2515-7647/ad2259

#### Glass corrosion: Mn K-edge µ-XANES/µ-XRF

Medieval glass, can undergo corrosion due to the formation of dark-coloured manganese-rich inclusions that reduce the transparency of the glass.

the presence of water and oxygen, Mn(II) and/or Mn (III) ions can be oxidised to higher oxidation states giving rise to, for example, insoluble Mn(IV) compounds (e.g., MnO<sub>2</sub>) that can precipitate, a process favoured in alkaline conditions.

Two different **restoration strategies** to improve the transparency: (i) reduction of highly oxidised black/brown compounds using mildly reducing solutions; (ii) extraction of manganese from the inclusions by the application of chelating agents.

Mn K-edge XANES/XRF mapping to evaluate the effect of reducing treatments on historical glass, dated to the 14th century and originating from Sidney Sussex College (Cambridge, UK), suffering from manganese browning.



#### Glass corrosion: Mn K-edge XANES spectroscopy\*



μ-XRF maps at selected energies for visualizing different oxidation states of manganese

#### Glass corrosion: Mn oxidation state mapping\*





**The hydroxylamine treatment** proved to be the **most effective in reducing the highly oxidised Mn compounds** in the inclusions, a change that could be observed visually and was supported by Mn oxitation state mapping.

Preliminary radiation damage tests: aimed at minimizing the photo-reduction of Mn(IV) to Mn(II) induced by the exposure to the X-ray beam.

5% w/w hydroxylamine

#### Manufacturing process of Roman ceramics\*

Black/red colour: the glossy black contains significant amounts of Fe(II), while the red ochre ground is composed almost entirely of Fe(III);

Produced from clays with very similar bulk chemistry but by manipulation of Fe oxidation state and mineralogy through skilled control of nanoscale porosity, kiln temperatures and oxidative environment.

> Study of two fragments of roman ceramics, from two different locations and times:

(i) <u>B type Campanian ceramic</u>: it has a high black gloss coating with a red non-vitrified body (found during the excavations of the old city of Narbonne and dated to the 1st century BCE).

(ii) Pre-sigillata fragment (La Graufesenque workshop; Millau, France - Augustan period): its paste is non-vitrified and has a red/brown colour while its slip is black. The slip is vitrified but not as brilliant as that of the Campanian ceramic.

Phase studies by full-filed XANES imaging at Fe K-edge



JAAS	RSCPublishing
PAPER	View Article Online View Journal  View Issue
Cite this: J. Anal. At. Spectrom., 2013, 28, 1870	Full-field XANES analysis of Roman ceramics to estimate firing conditions—A novel probe to study hierarchical heterogeneous materials
	Florian Meirer, <sup>a</sup> Yijin Liu, <sup>b</sup> Emeline Pouyet, <sup>c</sup> Barbara Fayard, <sup>c</sup> Marine Cotte, <sup>c</sup>

#### Manufacturing process of Roman ceramics\*: Full field-XANES imaging at Fe K-edge



#### **Pre-sigillata**

black slip

B type Campanian





Three mineral phases of Fe: hercynite [black, Fe(II)], hematite and maghemite [both red, Fe(III)]

➢ Very thin thicker and more uniform layer on the Pre-sigillata slip. Two possible explanations: (i) the surface layer was polished away after firing; (ii) the surface of the slip was protected from oxygen during reoxidation and the vessel oxidized from the inside out.

Pre-sigillata appears to be mostly hematite while that of the Campanian is mostly maghemite with a few large islands of hematite;

10 µm

**Red body** 

The final re-oxidation step of the three stage firing protocol (oxidative $\rightarrow$ reductive $\rightarrow$ oxidative) for the Campanian ceramic was performed under either lower oxygen fugacity or at significantly lower temperature and/or shorter duration than the Pre-sigillata ceramic.

## Stone materials – consolidation treatments



Materials Characterization 156 (2019) 109853

Inspecting adhesion and cohesion of protectives and consolidants in sandstones of architectural heritage by X-ray microscopy methods



Simona Raneri<sup>a,\*</sup>, Alessandra Giannoncelli<sup>b</sup>, Elisabeth Mascha<sup>c</sup>, Lucia Toniolo<sup>d</sup>, Marco Roveri<sup>d</sup>, Andrea Lazzeri<sup>e,f</sup>, Maria Beatrice Coltelli<sup>e,f</sup>, Luca Panariello<sup>e,f</sup>, Marco Lezzerini<sup>a</sup>, Johannes Weber<sup>c</sup>

#### Analytical Methods





# Ca speciation investigations for probing the carbonatation process of Ca-based consolidants into limestones\*

> Application of consolidants: common practice for the preservation of limestones used in archaeological/historical monuments against outdoor weathering.

> Recently, inorganic treatments involving the in situ formation of  $CaCO_3$  have been proposed, being more suitable due to their higher compatibility and stability with respect to synthetic polymers.

#### **RESEARCH PROBLEM**

> Main issue of Ca-based consolidants: formation of different CaCO<sub>3</sub> phases: **amorphous CaCO<sub>3</sub> (ACC)** deposits and crystalline systems composed of either **vaterite**, or calcite, or their mixtures.

> The formation of different phases of  $CaCO_3$  is relevant because it may have implications on the long-term performance of a consolidation treatment.

#### **MATERIALS and TREATMENTS**

Substrate: high porosity limestone (Lecce stone, with porosity of ~37%).

➤ Treated stones cured with calcium acetoacetate (CaAcAc) under different RH% conditions (50, 80 and 95%) and at 40°C.

> Analysis performed on "thin" cross-sections of treated substrate fragments (sizes of *ca.*  $2 \times 6 \text{ mm}^2$ ) embedded into resin and polished down to a thickness of *ca.* 150  $\mu$ m.



#### Why XANES spectroscopy at Ca K-edge?



#### Ca K-edge FF-XANES imaging\*





#### **EXPERIMENTAL ISSUE:**

Large (ca. 2×6 mm<sup>2</sup>) and heterogeneous samples.



#### Ca K-edge FF-XANES imaging/2D µ-XRD mapping\*



#### Ca K-edge FF-XANES imaging











2D μ-XRD mapping



# Lead calcium phosphate solid solutions: alteration compounds or original materials?

The presence of lead-calcium phosphates on artworks is not clear. It might be caused by:

- Intentional use as pigment or filler;
- A result of alteration processes of original materials.

Hydroxyapatite  $[Ca_5(PO_4)_3OH]$  can immobilize lead, thus producing a solid solution of  $Pb_xCa_{(5-x)}(PO_4)_3OH$  where Pb(II) ions mostly occupy Ca(II) sites.





Pyromorphite [Pb<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>(OH)] and/or Phosphohedyphane [(Pb,Ca)<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>CI]?

# Lead calcium phosphate solid solutions: synthesis and characterization

#### $5M^{2+} + 3PO_4^{3-} + OH^{-} => Pb_xCa_{(5-x)}(PO_4)_3OH$

M = Ca, Pb, (Ca+Pb)

( <sub>Pb</sub>	Samples	Formula	Ca:Pb Ratio
	HA	Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	1:0
	PbHA-10	(Pb <sub>0.1</sub> Ca <sub>0.9</sub> ) <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	0.9 : 0.1
	PbHA-20	(Pb <sub>0.2</sub> Ca <sub>0.8</sub> ) <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	0.8 : 0.2
	PbHA-40	(Pb <sub>0.4</sub> Ca <sub>0.6</sub> ) <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	0.4 : 0.6
	PbHA-50	(Pb <sub>0.5</sub> Ca <sub>0.5</sub> ) <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	0.5 : 0.5
	PbHA-70	(Pb <sub>0.7</sub> Ca <sub>0.3</sub> ) <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	0.3 : 0.7
	PbHA-80	(Pb <sub>0.8</sub> Ca <sub>0.2</sub> ) <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	0.2 : 0.8
	HPy	Pb <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	0:1

#### Analytical characterization of powders:

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- Morphological, microstructural, and local atomic structure: SEM, XRD, XANES spectroscopy at Ca K- and P K-edges.
- Intermediate for the structure: FT-IR and Raman spectroscopies.



Non-Destructive and Non-Invasive Approaches for the Identification of Hydroxy Lead–Calcium Phosphate Solid Solutions  $((Pb_xCa_{1-x})_5(PO_4)_3OH)$  in Cultural Heritage Materials



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#### Lead calcium phosphate solid solutions: IR, Raman & XRD\*





➤The substitution of lighter Ca<sup>2+</sup> ions with heavier Pb<sup>2+</sup> ions leads to the HA lattice expansion;

This is the main responsible for the changes observed in the position, shape, width, and intensity of  $v_3/v_4$  and  $v_1/v_2$  FT-IR and Raman bands (stretching/bending of phosphate group)

#### Lead calcium phosphate solid solutions: Ca K-/P K-edge XANES \*



White line and band shape and energy position of pre-/post-edge signals of the P K-/Ca K-edge XANES spectra suitable as markers for the non-destructive discrimination of different PbHA-X compounds at the microscale.

# Color changes in paintings: alteration of blue pigments

Fe K-and Co K-edges XAS analysis for the study of the fading of Prussian blue and discoloration of smalt

# Fading of Prussian blue

> **Prussian blue:** discovered at the beginning of  $18^{th}$  century and of composition MFe<sup>III</sup>[Fe<sup>III</sup>(CN)<sub>6</sub>]·xH<sub>2</sub>O (with M= K<sup>+</sup>, NH<sub>4</sub><sup>+</sup> or Na<sup>+</sup>). The intense blue color arises from an intervalent electron transfer between the Fe(II) and Fe(III) oxidation states when light is absorbed at about 700 nm.

Artificially aged model samples before and after light exposure have been investigated using a combination of Fe K-edge XAS analysis and iron-57 Mössbauer spectroscopy for the understanding of the alteration mechanism of this pigment.<sup>(a)</sup>

![](_page_25_Figure_3.jpeg)

View of Lake Sortedam from Dosseringen Looking Towards the Suburb Nørrebro outside Copenhagen (1838, Christen Købke; Statens Museum for Kunst, Copenhagen, DK).<sup>(b)</sup>

<sup>(a)</sup> L. Samain et *al.,* J. Anal. At. Spectrom. 26 (2011) 930-941; L. Samain *et al.,* J. Anal. At. Spectrom. 28 (2013) 524–535. <sup>(b)</sup> A. Vila *et al.,* in: "Science and Art: The Painted Surface" (Eds: A. Sgamellotti, B. G. Brunetti, C. Miliani), The Royal Society of Chemistry, London, 2014, 354-372.

# Fe K-edge XAS analysis\*

![](_page_26_Figure_1.jpeg)

![](_page_26_Figure_2.jpeg)

> **Fading** due to a  $Fe(III) \rightarrow Fe(II)$  reduction (in the XAS spectrum: decreasing of the intensity of the white line and shift of the absorption edge towards lower energies).

Disruption of the intervalent electron transfer Fe(II)-C-Fe(III).

Decrease in the Fe coordination number in the second and third shell that lead to a weakening of the intervalent Fe(II)/Fe(III) charge transfer.

The fading is influenced by the nature and amount of the white pigment.

 $\Delta \mu$  = (XANES spectrum<sub>unaged</sub>-XANES spectrum<sub>aged</sub>): direct correlation with the total colour change ( $\Delta E^*$ )

\* L. Samain et al., J. Anal. At. Spectrom. 26 (2011) 930-941.

## Smalt discoloration

> **Smalt** : blue pigment commonly used by artists between the  $16^{th}$  and  $18^{th}$  centuries. It is a potash glass in which the color stems from cobalt ions in the divalent oxidation state.

>XAS investigations of micro-samples taken from paintings and artificially aged model samples have established the alteration mechanism of this pigment.

![](_page_27_Picture_3.jpeg)

AAS J. Anal. At. Spectrom., 2012, 27, 1941–1948

# **Discoloration of the smalt pigment: experimental studies and** *ab initio* **calculations<sup>†</sup>**

Ilaria Cianchetta,<sup>*a*</sup> Ivan Colantoni,<sup>*b*</sup> Fabio Talarico,<sup>*c*</sup> Francesco d'Acapito,<sup>*d*</sup> Angela Trapananti,<sup>*d*</sup> Chiara Maurizio,<sup>*e*</sup> Simona Fantacci<sup>*f*</sup> and Ivan Davoli<sup>*b*</sup>

# Co K-edge XAS (XANES/EXAFS) analysis\*

![](_page_28_Figure_1.jpeg)

> **Discoloration** due to a change in the local environment of  $Co^{2+}$  from tetrahedral (blue) to octahedral (pink) while the leaching of K<sup>+</sup> takes place. The process does not involve modifications of the oxidation state of the cobalt ion.

\*Figures re-adapted from L. Robinet et al., Anal. Chem. 83 (2011) 5145–5152.

# Color changes in paintings: alteration of yellow pigments

Cr K-edge XAS analysis for the study of the darkening of chrome yellows

### Darkening of chrome yellows (PbCr<sub>1-x</sub>S<sub>x</sub>O<sub>4</sub>)

> Characterized by low photochemical stability with tendency to lose their original brilliant yellow color.

![](_page_30_Picture_2.jpeg)

\*L. Monico *et al., Anal. Chem.* 83 (2011) 1214–1223; L. Monico *et al., Anal. Chem.* 83 (2011) 1224-1231; L. Monico *et al., Anal. Chem.* 85 (2013) 860-867; L. Monico *et al., Anal. Chem.* 86 (2014) 10804–10811; L. Monico *et al., JAAS* 30 (2015) 613-626; L. Monico *et al., JAAS* 30 (2015) 1500-1510; L. Monico *et al., Angew. Chem. Int. Ed.* 54 (2015) 13923-13927; L. Monico *et al., ACS Omega* 4 (2019) 6607-6619.

#### Cr speciation investigations: Cr K-edge XANES

![](_page_31_Figure_1.jpeg)

**Cr(VI)** compounds: non-centrosymmetric tetrahedral coordination.

>Cr(III) compounds: centrosymmetric octahedral geometry.

> Pre-edge peak area proportional to the relative amount of Cr(VI).

Shift of the absorption edge position towards higher energies: *increase of the valency* of the absorbing atom and/or of the *electronegativity* of the nearest neighbour atoms.

>Identification of specific reduced Cr-compounds challenging, when different Cr-species are co-present.

![](_page_32_Picture_0.jpeg)

![](_page_32_Picture_1.jpeg)

# Van Gogh's Sunflowers (Amsterdam version)

Angewandte Chemie

Evidence for Degradation of the Chrome Yellows in Van Gogh's Sunflowers: A Study Using Noninvasive In Situ Methods and Synchrotron-Radiation-Based X-ray Techniques

Letizia Monico,\* Koen Janssens, Ella Hendriks, Frederik Vanmeert, Geert Van der Snickt, Marine Cotte, Gerald Falkenberg, Brunetto Giovanni Brunetti, and Costanza Miliani

Angew. Chem. 2015, 127, 14129-14133

![](_page_32_Picture_7.jpeg)

Chemical Mapping by Macroscopic X-ray Powder Diffraction (MA-XRPD) of Van Gogh's *Sunflowers*: Identification of Areas with Higher Degradation Risk

Frederik Vanmeert,\* Ella Hendriks, Geert Van der Snickt, Letizia Monico, Joris Dik, and Koen Janssens

Angew. Chem. Int. Ed. 2018, 57, 7418-7422

![](_page_32_Picture_12.jpeg)

Amsterdam University Press

#### Cr speciation analyses: aged paint mock-ups & original paint micro-samples

![](_page_33_Figure_1.jpeg)

![](_page_33_Picture_2.jpeg)

Thin section (thickness: ~5 µm)

![](_page_33_Figure_4.jpeg)

step size (h×v): 0.8×0.3 µm<sup>2</sup> dwell time: 100 ms/pixel Energy: 5.993-6.1 keV

![](_page_33_Figure_6.jpeg)

![](_page_33_Picture_7.jpeg)

![](_page_33_Picture_8.jpeg)

dwell time: 100 ms/pixel Energy: 5.993-6.1 keV

35

Sunflowers (F458, January 1889, Van Gogh Museum)

\*L. Monico et al., Anal. Chem. 85 (2013) 860-867; L. Monico et al., JAAS 30 (2015) 1500-1510; L. Monico et al., Angew. Chem. Int. Ed. 54 (2015) 13923-13927.

#### Maia and SSD-detector based microprobe systems: The Bedroom\*

![](_page_34_Figure_1.jpeg)

Microprob e system	Map size (h×v) (μ²)	Pixel size (h×v) (µm²)	Pixel total number	Dwell time (ms/pixel)	Acquisition time (min)	Absolute photon flux (ph/s)	fluence (ph/µm²)
Maia-P06	300×80	1×1	2.4×10 <sup>4</sup>	3	210	9×10 <sup>8</sup>	-8×10 <sup>8</sup>
Maia-XFM	300×80	1×1	2.4×10 <sup>4</sup>	0.99	130	2.1×10 <sup>8</sup>	-2.6×10 <sup>7</sup>
SDD-ID21	73.8×22.6	0.6×0.2	1.3899×10 <sup>4</sup>	150-300	108	1.7×10 <sup>8</sup>	~3.3×10 <sup>7</sup>

\* L. Monico et al., JAAS 30 (2015) 613-626.

#### Cr K-edge XANES spectroscopy in total electron yield (TEY) & total X-ray fluorescence (XFY) modes

![](_page_35_Figure_1.jpeg)

<sup>\*</sup> L. Monico et al., Nuclear Instruments and Methods in Physics Research B 539 (2023)141–147.

# Radiation damage on cultural heritage materials: further readings

![](_page_36_Picture_1.jpeg)

Loïc Bertrand <sup>a,b,\*</sup>, Sebastian Schöeder <sup>b</sup>, Demetrios Anglos <sup>c,d</sup>, Mark B.H. Breese <sup>e</sup>, Koen Janssens <sup>f</sup>, Mehdi Moini <sup>g</sup>, Aliz Simon <sup>h,i</sup>

#### Trends in Analytical Chemistry 164 (2023) 117078

![](_page_36_Picture_4.jpeg)

Practical advances towards safer analysis of heritage samples and objects

![](_page_36_Picture_6.jpeg)

Loïc Bertrand <sup>a, \*</sup>, Sebastian Schöder <sup>b</sup>, Ineke Joosten <sup>c</sup>, Samuel M. Webb <sup>d</sup>, Mathieu Thoury <sup>e</sup>, Thomas Calligaro <sup>f</sup>, Étienne Anheim <sup>g</sup>, Aliz Simon <sup>h</sup>

# Safe analysis conditions of chrome yellows by SR X-ray methods

![](_page_37_Picture_1.jpeg)

#### Cr-Kβ XES/ Cr K-edge HERFD-XANES (ID26)\*

![](_page_38_Figure_1.jpeg)

#### Cr-Kβ XES/ Cr K-edge HERFD-XANES (ID26)\*

![](_page_39_Figure_1.jpeg)

### Cr-speciation mapping (ID21)\*

![](_page_40_Figure_1.jpeg)

### Cr K-edge µ-XANES (ID21): effect of S content\*

![](_page_41_Figure_1.jpeg)

<sup>\*</sup>L. Monico et al., Anal. Chem. 92 (2020) 14164-14173.

## Cr K-edge µ-XANES (ID21): effect of the binder\*

![](_page_42_Figure_1.jpeg)

<sup>\*</sup>L. Monico et al., Anal. Chem. 92 (2020) 14164-14173.

## µ-XRD (P06-PETRA III)\*

![](_page_43_Figure_1.jpeg)

Formation of an amorphous phase (clearly visible for PbCrO<sub>4</sub>-oil);

Loss of crystalline structure (decreasing of the intensity of the diffraction signals);

It is more pronounced for the sulfate-richer phases.

#### Cr-speciation mapping

![](_page_43_Picture_6.jpeg)

# Mitigation strategies: optimization of the fluence/dose & vacuum conditions \*

Different extent of photo-induced reduction for similar doses (different fluence/dose threshold);

Beamline	Energy (keV)	Fluence threshold (ph/µm²)	Absorbed dose threshold (MGy)	Vacuum
ESRF-ID21	~6	~5×10 <sup>11</sup>	~2×10 <sup>4</sup>	Yes
ESRF-ID26	~6	~10 <sup>8</sup>	~10	No
DESY-P06	21	~1-2×10 <sup>11</sup>	~2-4×10 <sup>3</sup>	No

Adapt time to stay below the established threshold value (fast-data acquisition – ID26 beamline);

- Decreasing the flux of the incoming beam (e.g., using attenuators of different thickness) as long as an adequate signal-to-noise-ratio is maintained;
- Defocusing the beam to minimize the fluence/dose to the sample, but sometimes at the expense of spectral resolution (e.g., XES analysis).
- Vacuum conditions: explain the lower Cr<sup>III</sup>-abundances obtained at ESRF-ID21;
- Such sample environment may contribute to indirectly slowing down the Cr-reduction due to the absence/neglectable content of air gases (e.g., O<sub>2</sub>) and moisture, that favor the oxidative degradation of the binding medium.
- \*L. Monico et al., Anal. Chem. 92 (2020) 14164-14173.

#### Mitigation strategies: lowering the temperature\*

![](_page_45_Figure_1.jpeg)

<sup>\*</sup>L. Monico et al., Anal. Chem. 92 (2020) 14164-14173.

![](_page_46_Figure_1.jpeg)

First ring: partners of the initial BAG proposal Second ring: indirect partners through collaborations Access to μ-XRD mapping (ID13, 4 days) HR-XRD (ID22, 3 days) every 6 months for 2 years

Beamtime shared between the partners Synergy and mutual support A very good training programme

https://www.esrf.fr/BAG/HG172

Heritage-bag@esrf.fr

![](_page_46_Picture_7.jpeg)

![](_page_46_Picture_8.jpeg)

![](_page_46_Picture_9.jpeg)

#### FROM STANDARD ACCESS MODELS TO BLOCK ALLOCATION GROUP (BAG) ACCESS

![](_page_47_Picture_1.jpeg)

![](_page_47_Picture_2.jpeg)

![](_page_47_Picture_3.jpeg)

Inspired from

proteins crystallography access mode